

# Polymers in Solution

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Dresden, 9<sup>th</sup> November 2022

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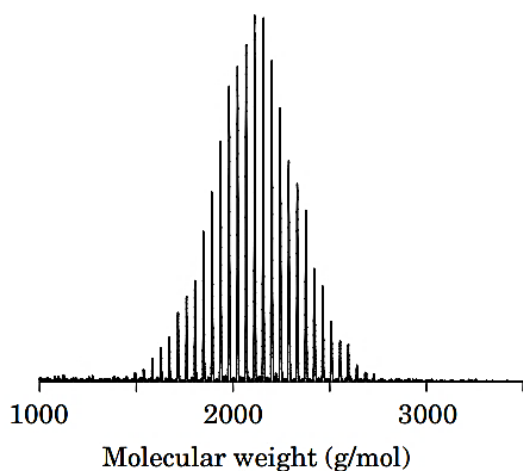
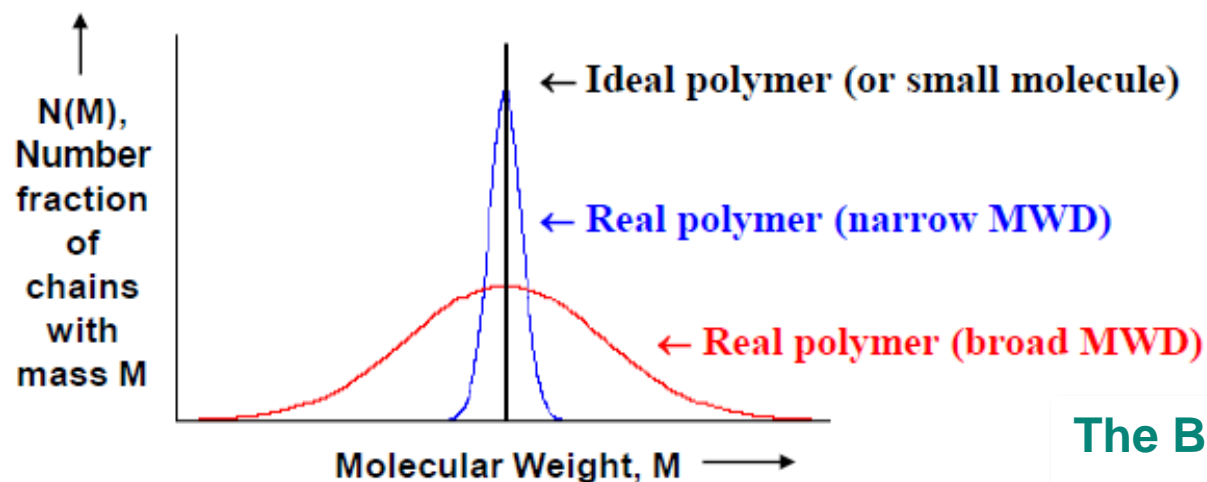
Bioactive and Responsive Polymers  
Institute of Macromolecular Chemistry  
Leibniz-Institut für Polymerforschung Dresden e.V.

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## **4. The statistical character of macromolecules**

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## 1. Introduction



**Figure 1.54.** Example of a mass spectrum of a polymer obtained in MALDI-TOF mass spectrometry. Poly(ethylene glycol) of a nominal molecular weight of 2,000 g/mol is shown. (From Ref. 10.)

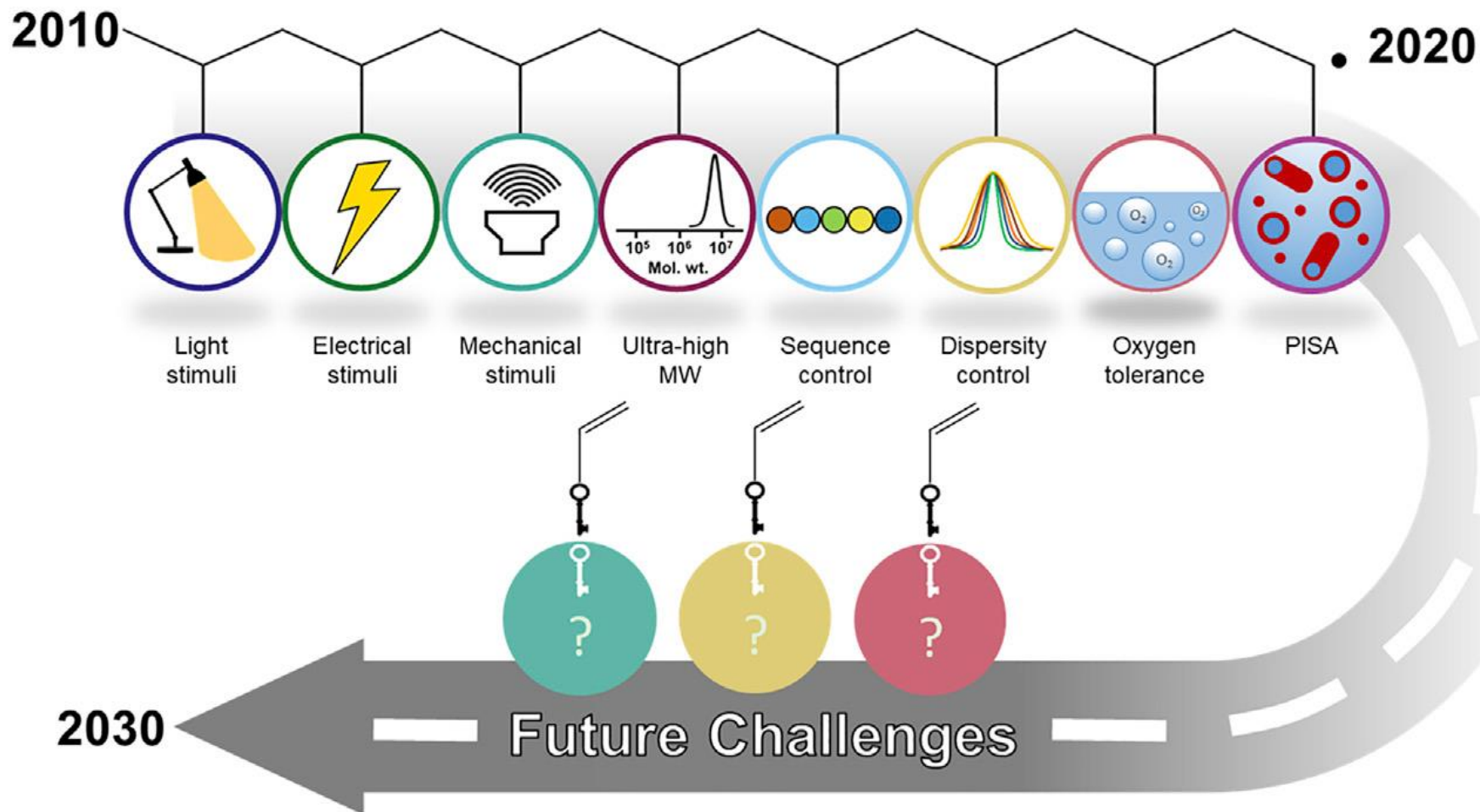
## The Bigger Picture: Challenges and opportunities

Universal catalysts compatible with a range of monomers and stimuli are urgently required to produce materials **with enhanced control over monomer sequence, dispersity, and tacticity**

New **depolymerization** and **degradations** for polymer recycling and bio-relevant applications should be the focus of future research

**Advanced characterization methods, automation systems,** and machine learning are currently undeveloped and could offer exciting new avenues

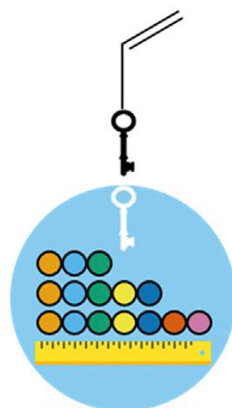
# Controlled Radical Polymerization



# Future Challenges



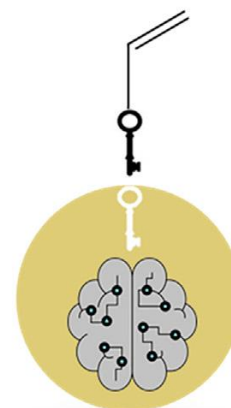
*In-situ*  
monitoring



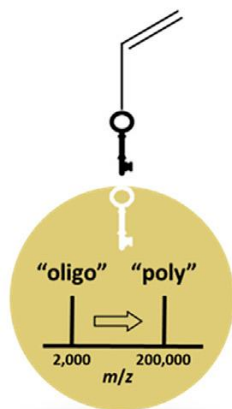
Size and  $D$   
characterization



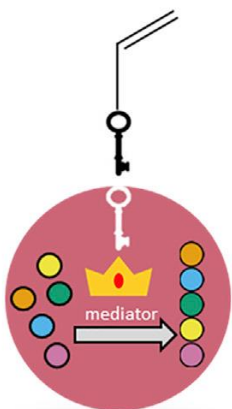
Automation



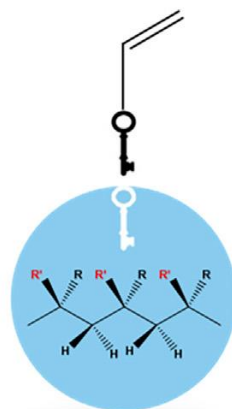
Machine  
learning



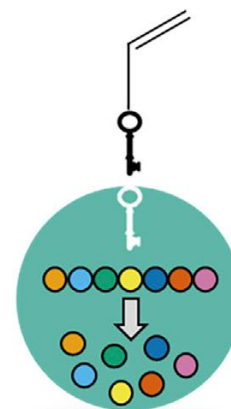
Monodisperse  
polymers



Universal CRP



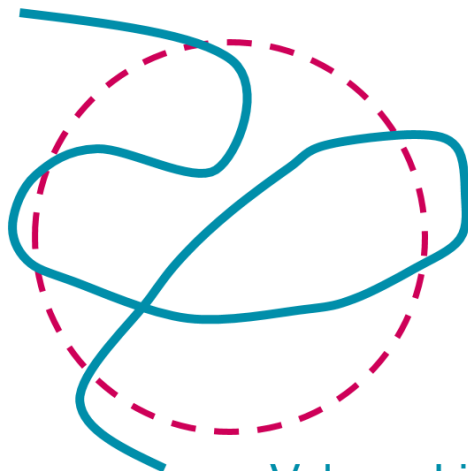
Tacticity control



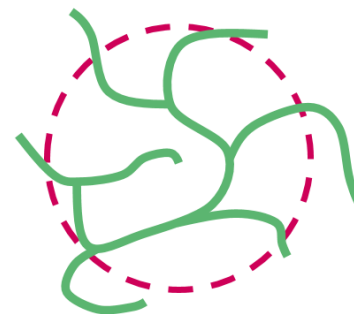
Depolymerization

## 2. Polymers with different degree of branching

### Linear Chain



### Branched Chain



At Equal MW

Volume Linear > Volume Branched

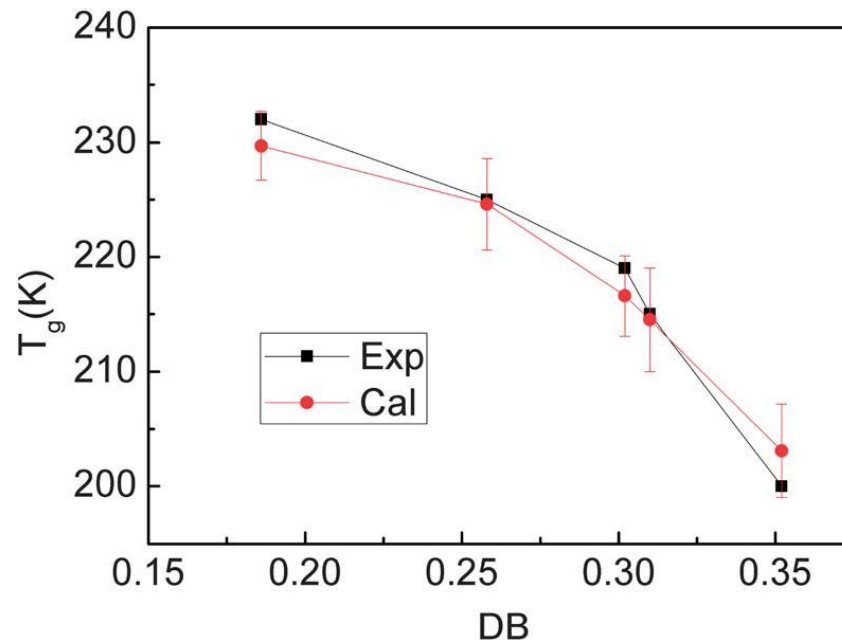
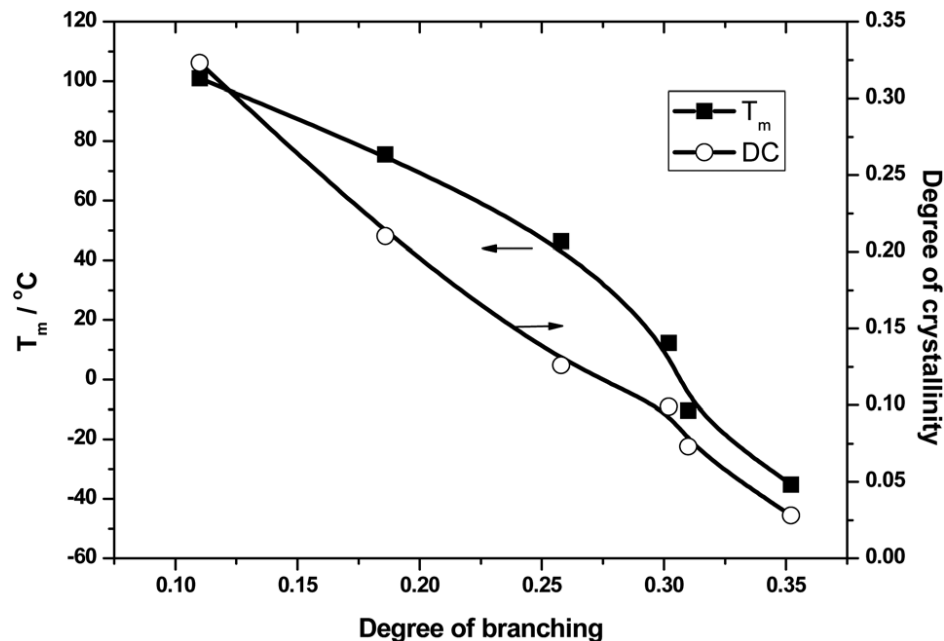
Molecular Density Linear < Molecular Density Branched

Intrinsic Viscosity Linear > Intrinsic Viscosity Branched

I	II	III	IV			
Linear	Crosslinked	Branched	Dendritic			
1930's	1940's	1960's		1980's		

## 2. Polymers with different degree of branching

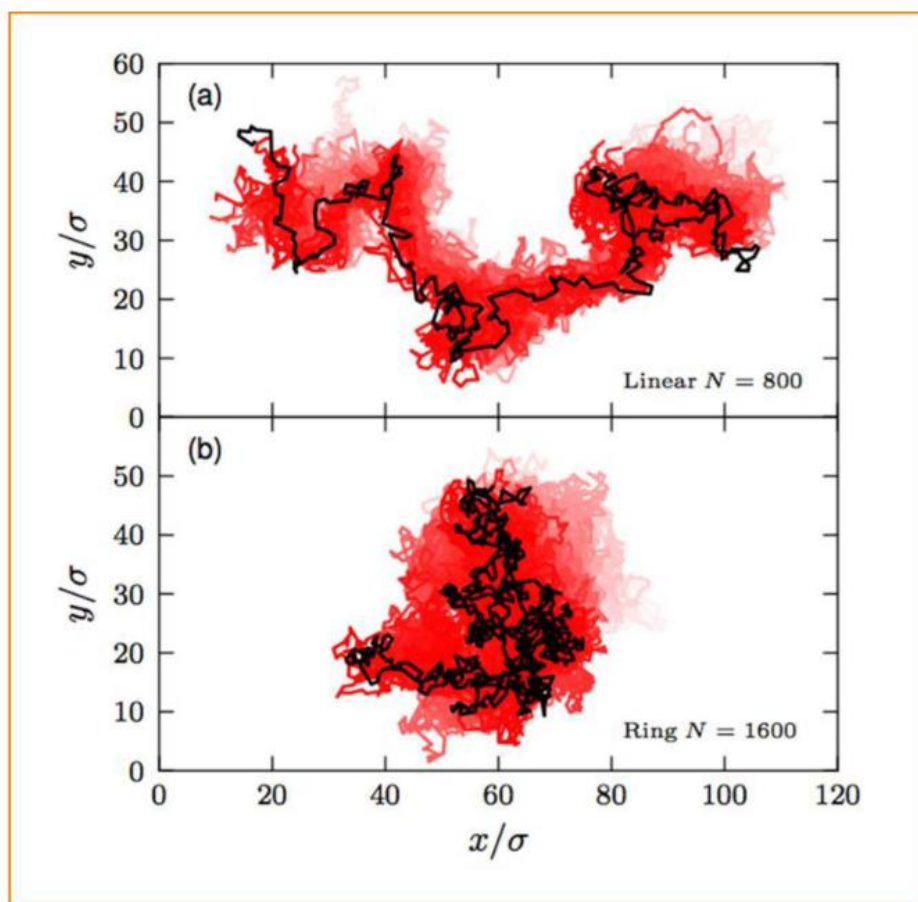
The relationship between the degree of branching and glass transition temperature of branched polyethylene: experiment and simulation



$$DB = \frac{D + T}{D + T + L}$$

## 2. Polymers with different degree of branching

## Polymer Topology Matters



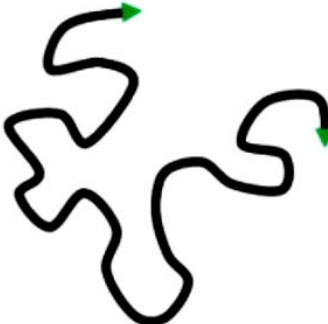
- Linear chains move like a snake

- Rings move by an amoeba-like motion



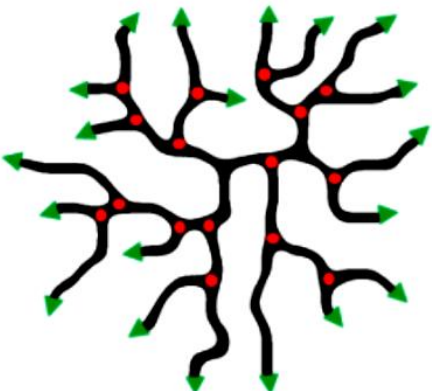
## 2. Polymers with different degree of branching

**linear polymer**



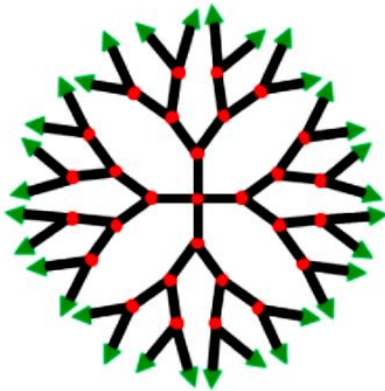
linear  
one-step synthesis  
easy purification by precipitation  
easy scale-up  
dispersity ( $\bar{D}$ ) > 1.01  
**DB = 0**  
no cavity  
2 end-groups  
strong entanglements  
high viscosity

**hyperbranched polymer**



three-dimensional, irregular  
one-step synthesis  
easy purification by precipitation  
easy scale-up  
 $\bar{D} > 1.1$   
**DB = 0.4-0.6**  
presence of cavities  
multiple end-groups  
weak entanglements  
low viscosity

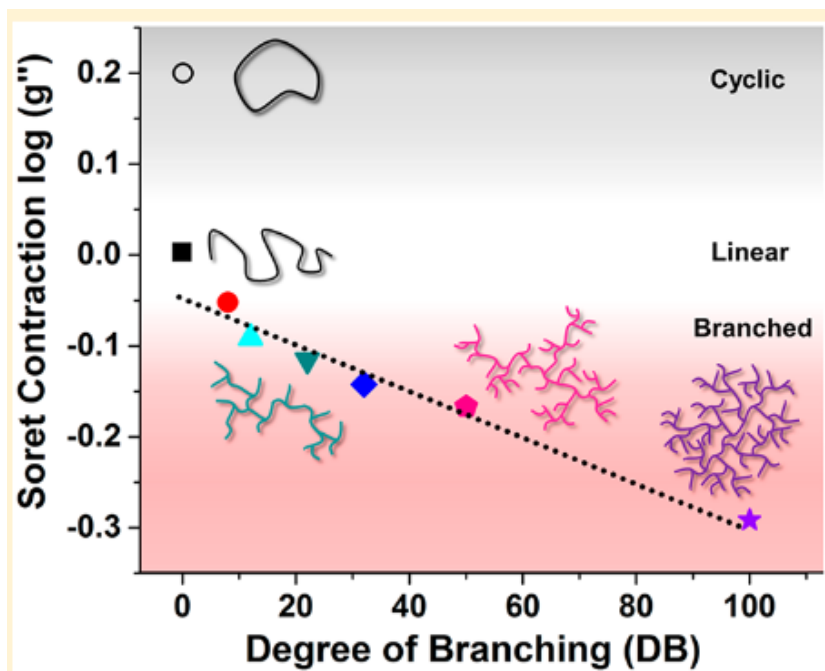
**dendrimer**



three-dimensional, regular  
multi-step synthesis  
purification by chromatography  
difficult scale-up  
 $\bar{D} = 1.0$   
**DB = 1.0**  
numerous cavities  
large number of end-groups  
no entanglements  
very low viscosity

## 2. Polymers with different degree of branching

Thermal field-flow fractionation (ThFFF) was used to characterize the architecture of aromatic–aliphatic polyesters with varying degrees of branching

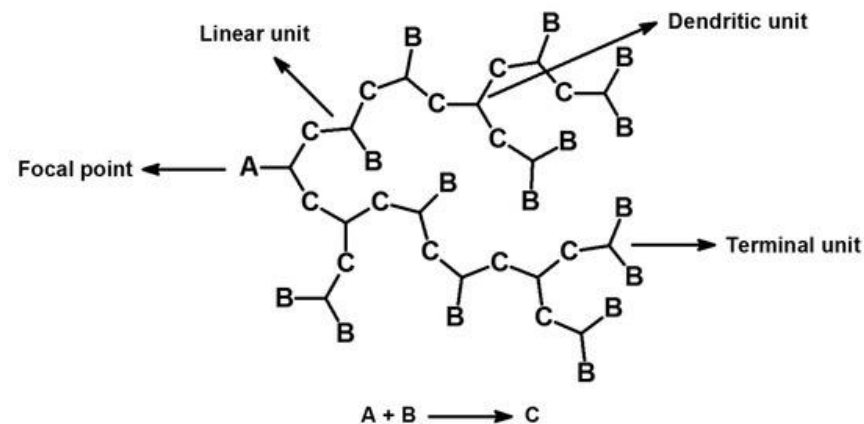


Thermophoretic force

$$S_T = D_T/D$$

$D$ , diffusion coefficient;  $D_T$  the thermodiffusion coefficient. The quotient of both coefficients is called

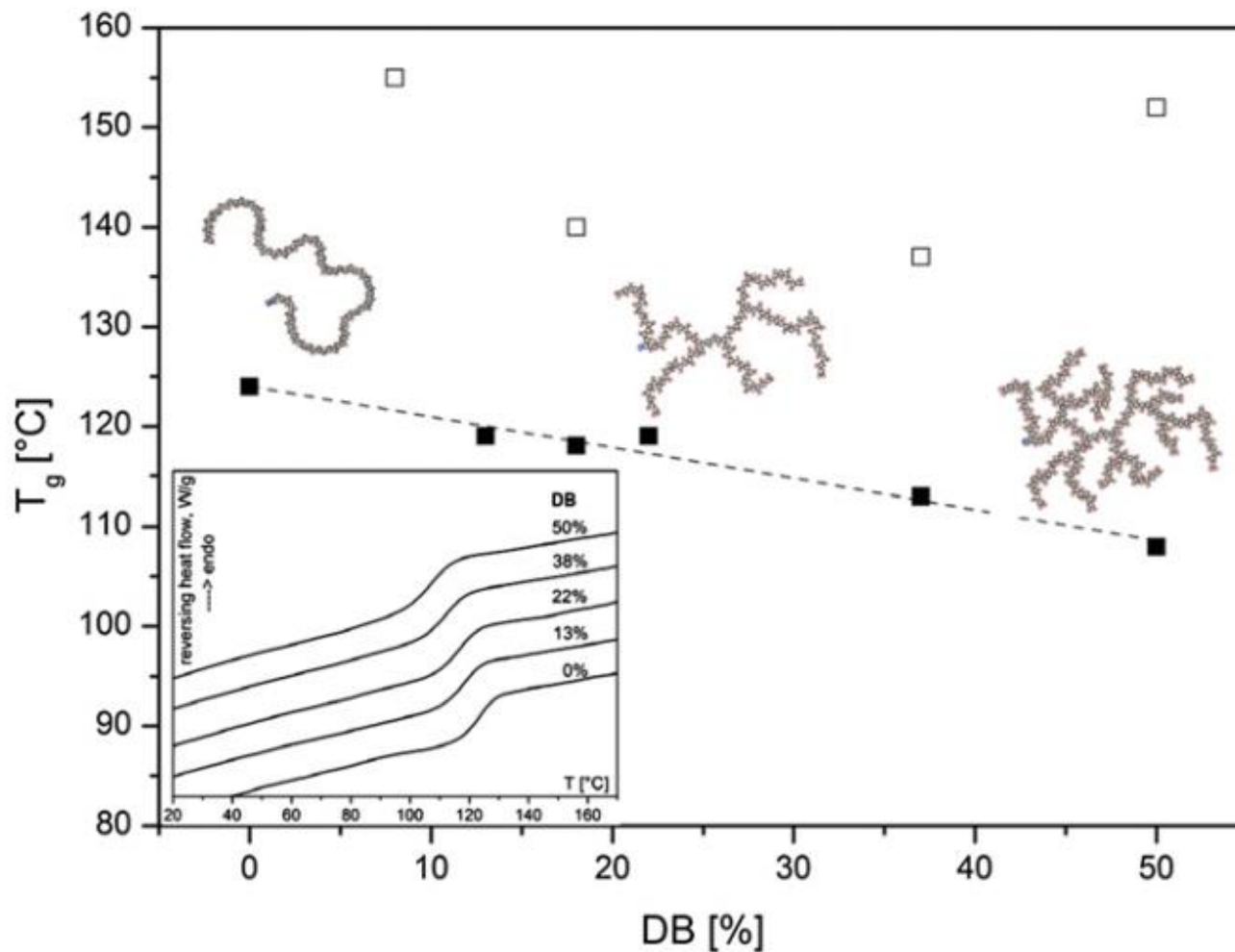
**Soret coefficient ( $S_T$ ).**



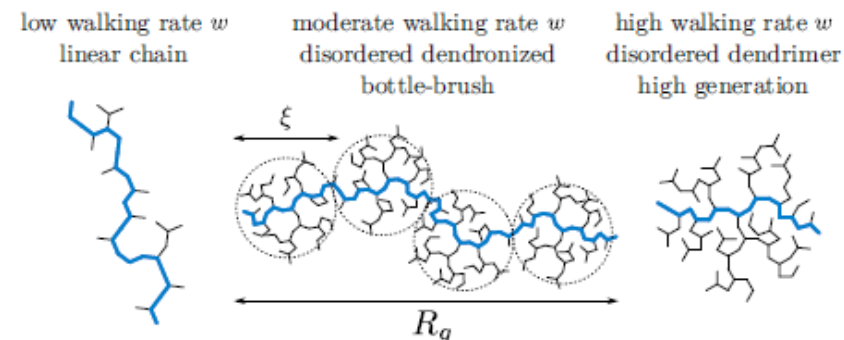
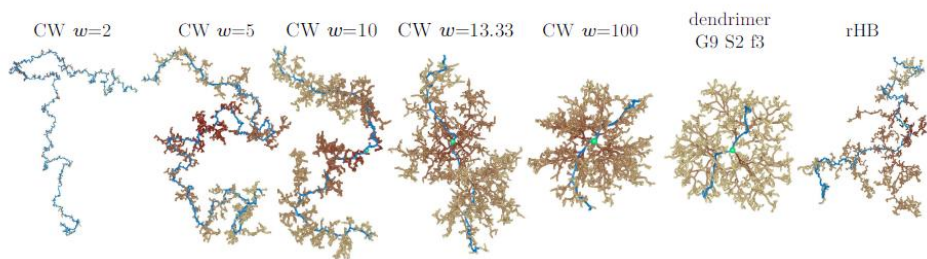
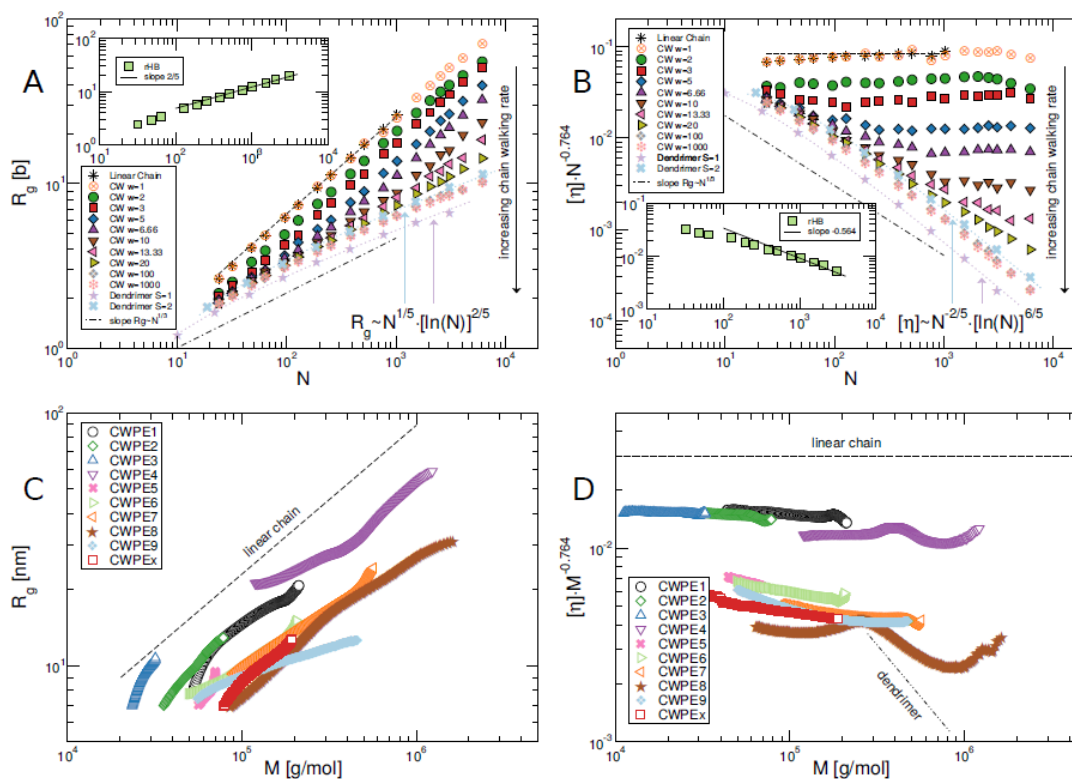
A **Soret contraction factor ( $g''$ )**, defined as the ratio of the ST of a branched polymer to the ST of a molecular weight equivalent linear analogue, is introduced **as a metric to indicate degree of branching (DB)**.

## 2. Polymers with different degree of branching

Effect of the degree of branching on the glass transition temperature of polyesters

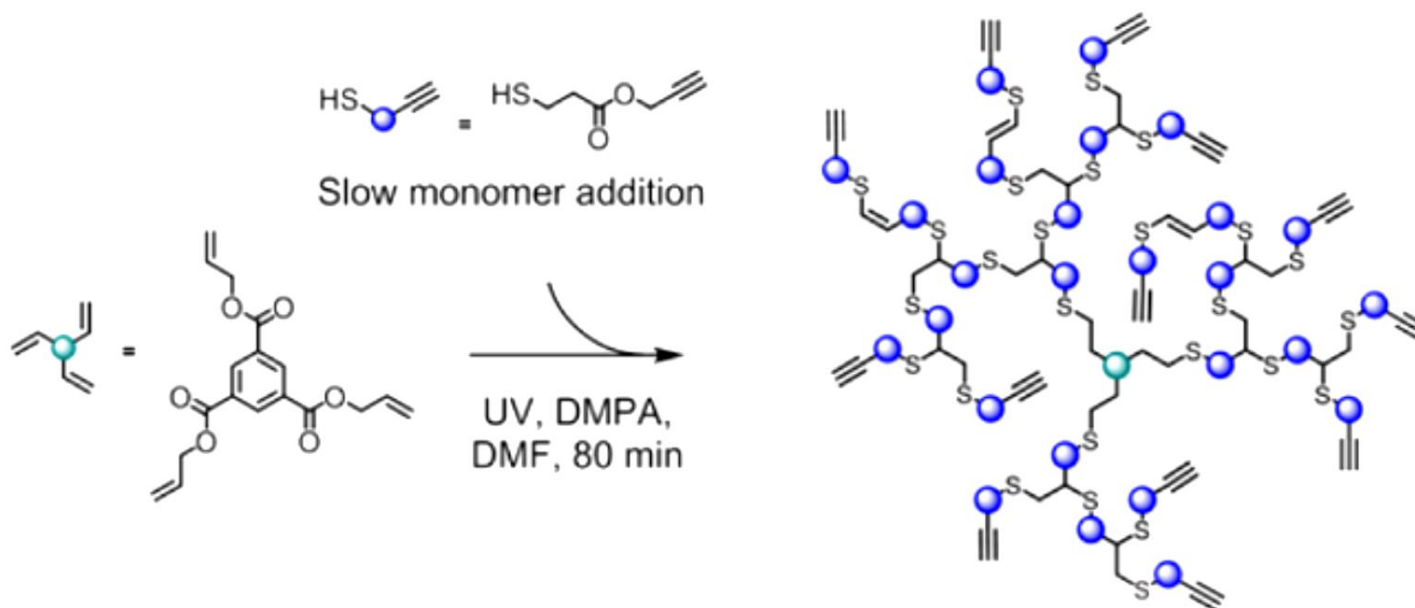


## 2. Polymers with different degree of branching



## 2. Polymers with different degree of branching

### Hyperbranched Polymers with High Degrees of Branching and Low Dispersity Values: Pushing the Limits of Thiol–Yne Chemistry



✓ Fast, UV, one-pot

✓ High degrees of branching

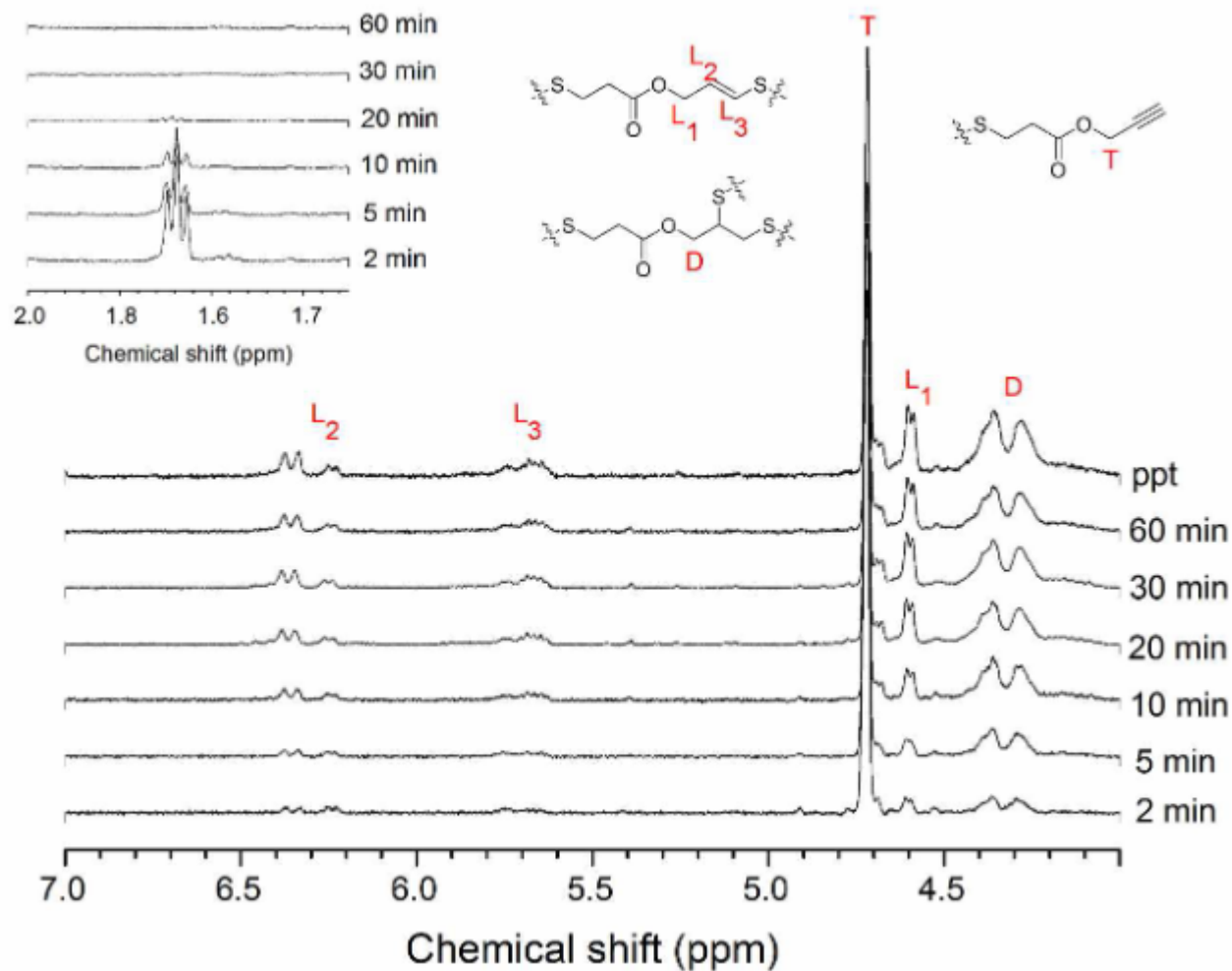
✓ Low dispersity values

Degrees of branching were determined by  $^1\text{H}$  NMR spectroscopy **to be greater than 0.8 in most cases.**

Increasing the fraction of core molecule was found to decrease dispersity to values as low as **1.26 and 1.38** for the alkene core and alkyne core.

## 2. Polymers with different degree of branching

### Hyperbranched Polymers with High Degrees of Branching and Low Dispersity Values: Pushing the Limits of Thiol–Yne Chemistry



$$DB = \frac{D + T}{D + T + L}$$

### 3. Polydispersity and Mechanism of Polymerization

Depends on the reaction mechanism of the polymer formation reaction and the chosen conditions (e.g. p, T)

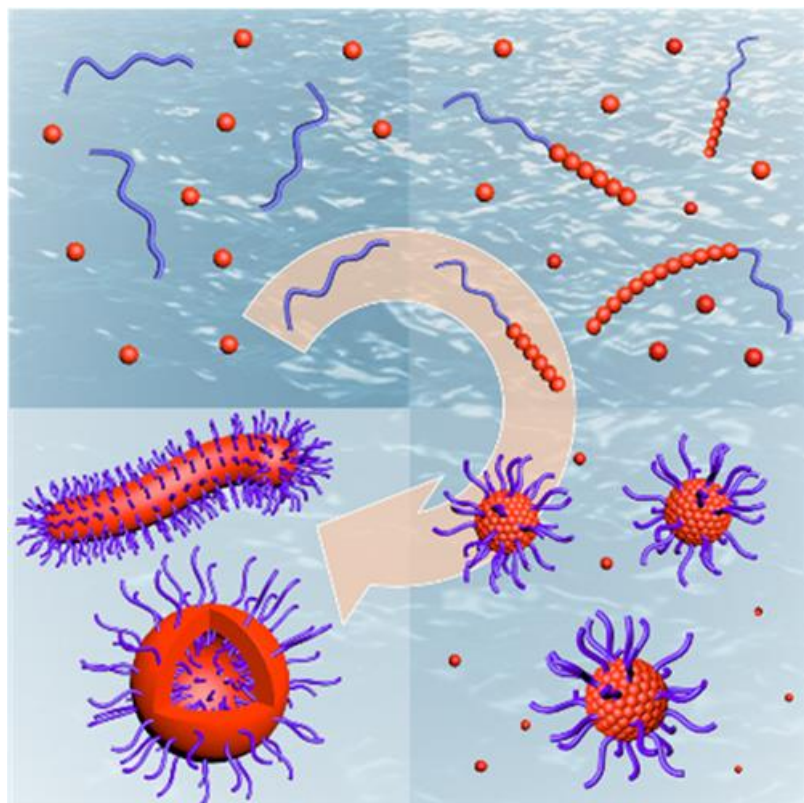
Kind of polymer	Reaction	$M_w/M_n$
Living Polymers	Anionic Group transfer	1.0...1.05
Condensation polymer	Step reaction Bifunctional Monomers	~ 2
Addition polymers	Radical addition	2-10
	Cationic addition Coordination polymerization (Organometallic complexes)	2-30
Branched Polymers	Radical	2-50
Networked Polymers	Step reaction of tri-, tetrafunctional Monomers	$\infty$ at the gel point

**Atom transfer radical polymerisation (ATRP), reversible addition–fragmentation chain-transfer (RAFT) polymerisation and nitroxide-mediated radical polymerisation (NMP)** have enabled the synthesis of well-defined macromolecules with controlled molecular weight, architecture, end-group fidelity and dispersity. **Dispersities in the range of  $\bar{D} = 1.01–1.20$**



### 3. Polydispersity and Mechanism of Polymerization

#### Principles and Characteristics of Polymerization-Induced Self-Assembly (PISA) with Various Polymerization Techniques



Preparation of a pre-defined polymer architecture is a key challenge to a researcher.

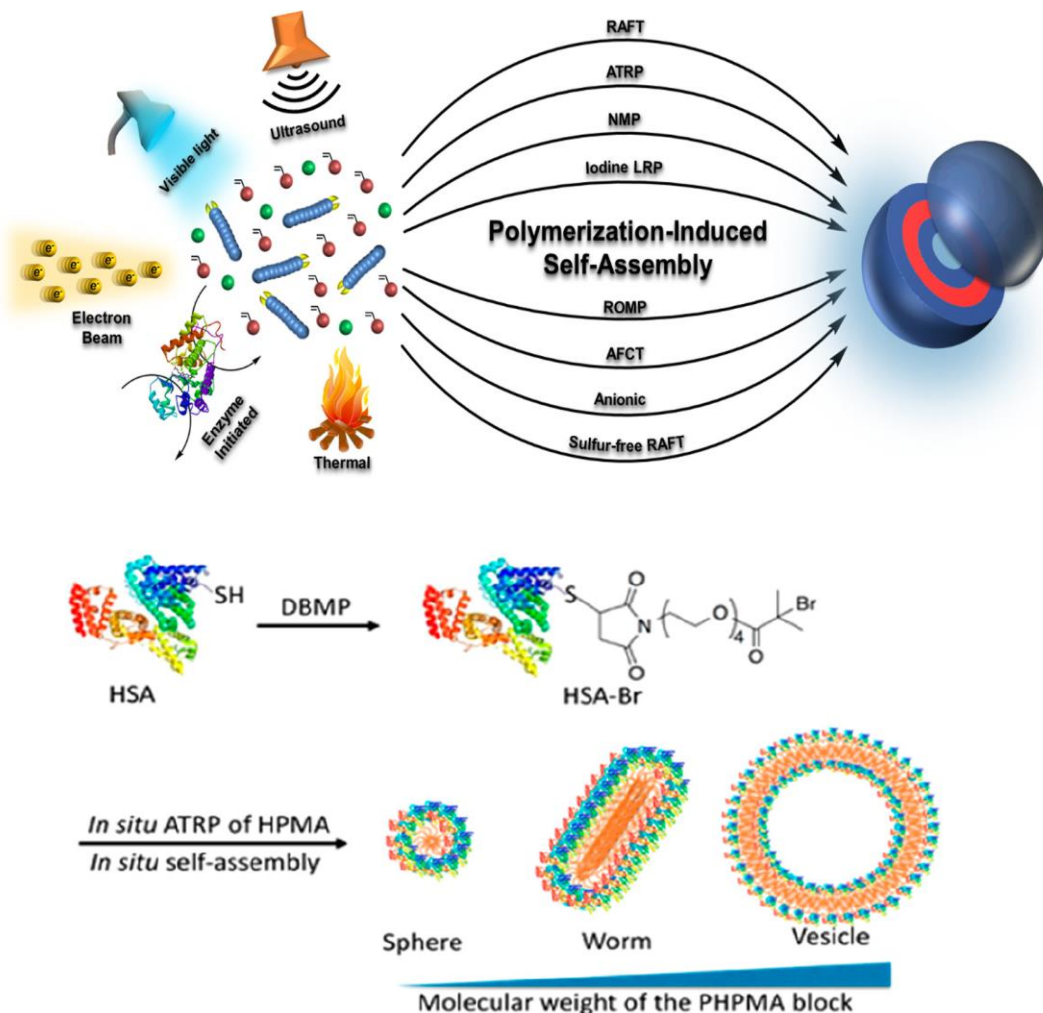
Polymerization Induced Self-Assembly (PISA) to synthesize polymeric nanoparticles with **pre-defined morphology and precise control over size and shape**. This methodology has become a potential strategy for the synthesis of various **block copolymer nano-objects**. The PISA strategy produces core-shell polymeric nanoparticles with a wide scope of morphologies including **spheres, worms, rods, and vesicles**.

A few parameters including the **degree of polymerization, core-forming monomers, macro-CTA and solid content of the final product** are precisely employed for the PISA procedure to accomplish the ideal size and shape.



### 3. Polydispersity and Mechanism of Polymerization

#### Principles and Characteristics of Polymerization-Induced Self-Assembly (PISA) with Various Polymerization Techniques



**Alternative PISA protocols**, which allow the preparation of nanoparticles with improved control over copolymer morphology and functionality.

For example, initiation based on visible light, redox chemistry, or enzymes enables the incorporation of sensitive monomers and fragile biomolecules into block copolymer nanoparticles.

## 4. Distributions. Statistical growth

The statistical growth of a constant number of chains in a living polymerization leads to the narrow Poisson distribution of the mole fraction as a function of the degree of polymerization  $N$ :

$$x(N) = \frac{v^{N-1} e^{-v}}{\Gamma(N)}$$

$v = \langle N \rangle_n - 1$  and  $\Gamma(N)$  is the gamma function. The corresponding distribution in terms of mass fractions is

$$w(N) = \frac{N v^{N-1} e^{-v}}{\Gamma(N)(v+1)}$$

The polydispersity decreases with increasing degree of polymerization and depends only on  $\langle N \rangle_n$ :

$$\frac{\langle N \rangle_w}{\langle N \rangle_n} = 1 + \langle N \rangle_n^{-1} - \langle N \rangle_n^{-2} \approx 1 + \langle N \rangle_n^{-1}$$

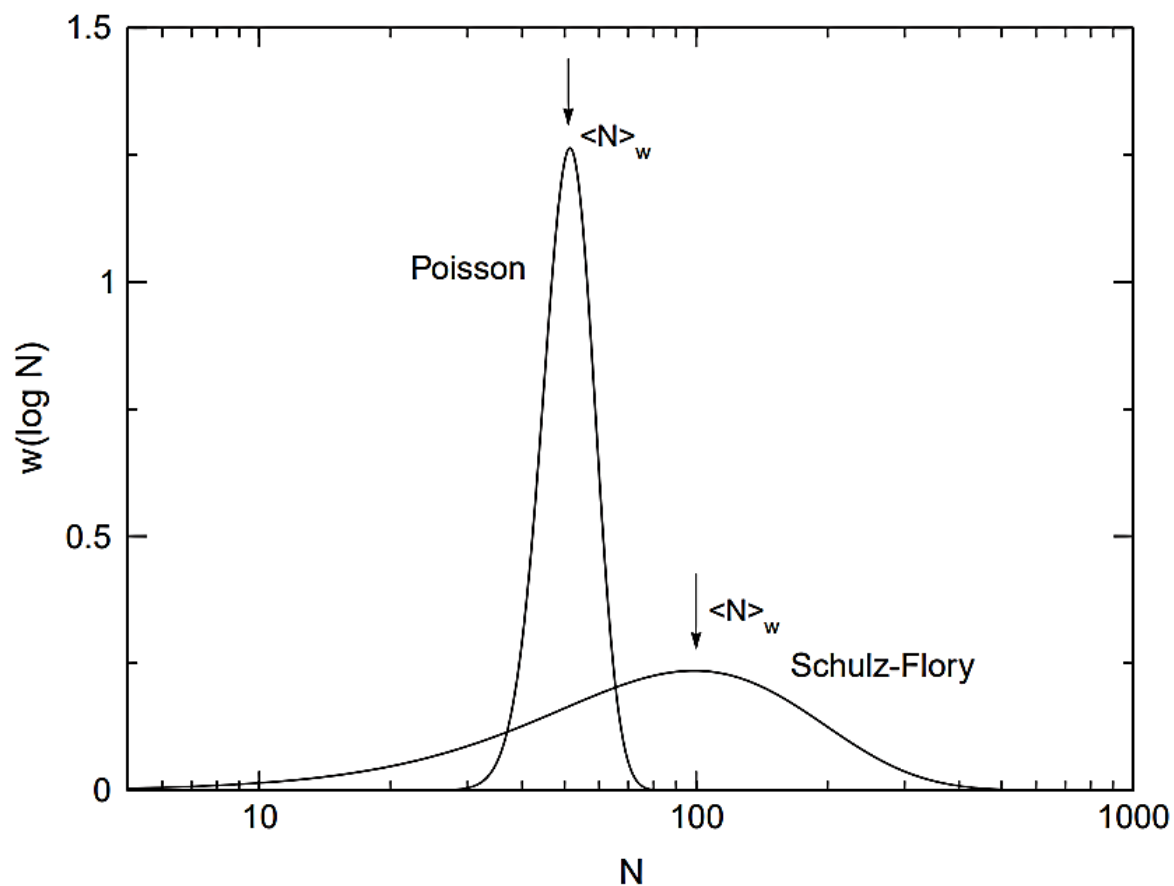
The much broader Schulz-Flory distribution

$$w(N) = \langle N \rangle_n^{-2} N (1 - \langle N \rangle_n^{-1})^{N-1}$$

with a constant polydispersity of  $\langle N \rangle_w / \langle N \rangle_n = 2$  is known from radical polymerization [9].

## 4. Distributions. Statistical growth

## Typical Distributions



Poisson and Schulz-Flory distribution with identical  $N = 50$ . The arrows indicate  $N_w = 51$  (Poisson) and  $N_w = 100$  (Schulz-Flory).

## 4. Distributions. Statistical growth

### Step-growth Polymerization

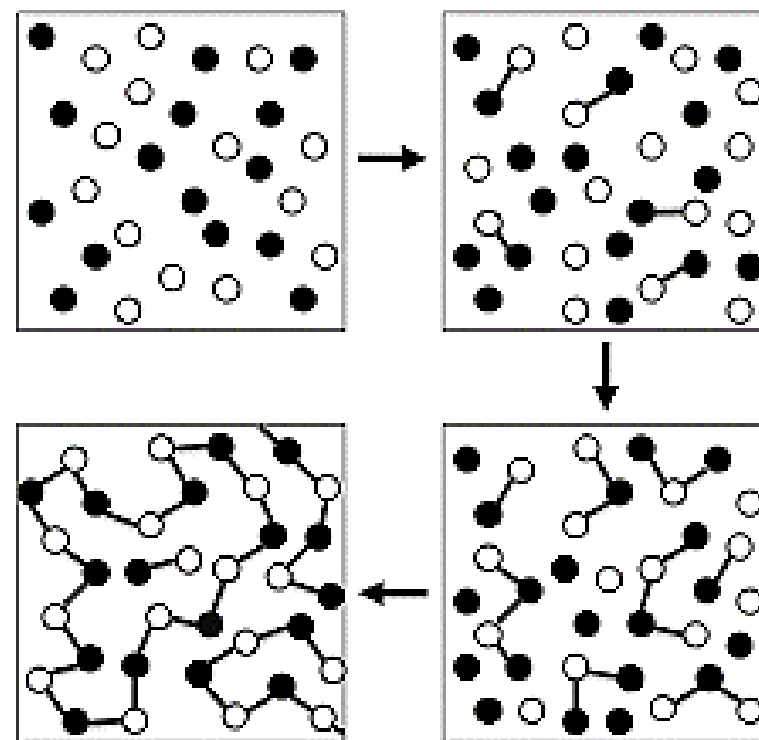
Step growth polymerization is the formation of a polymer from bi-functional or multifunctional monomers.

Self-condensation of A-B and the stoichiometric polymerization of A-A with B-B where A may react only with B and vice-versa.

Let  $p$  = probability that a B group has reacted (This is equivalent to the fraction of B groups reacted)

$1 - p$  = probability that a B group is unreacted

In virtually all cases one can assume that the reaction events are independent. Thus, the probability that an  $x$ -mer has formed is given by  $p^{x-1}(1-p)$

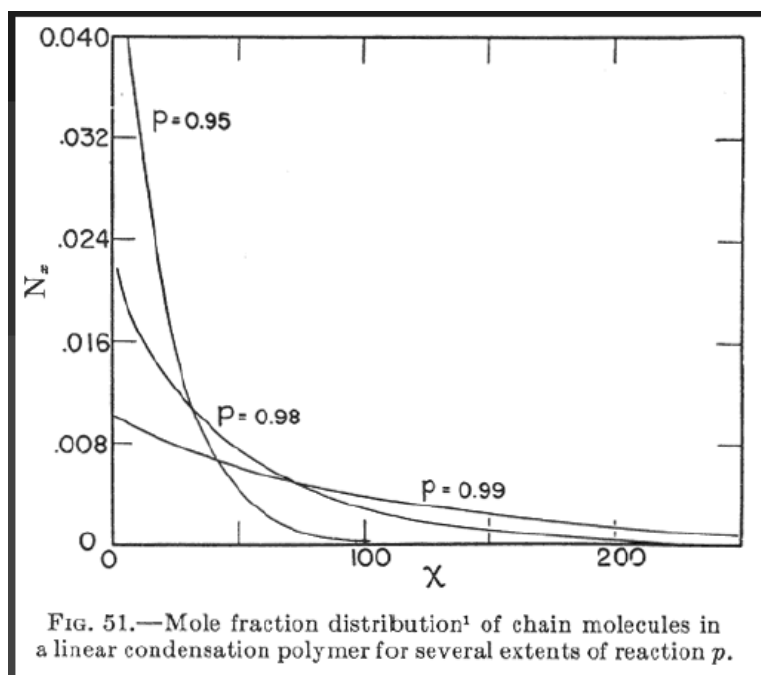


## 4. Distributions. Statistical growth

## Step-growth Polymerization

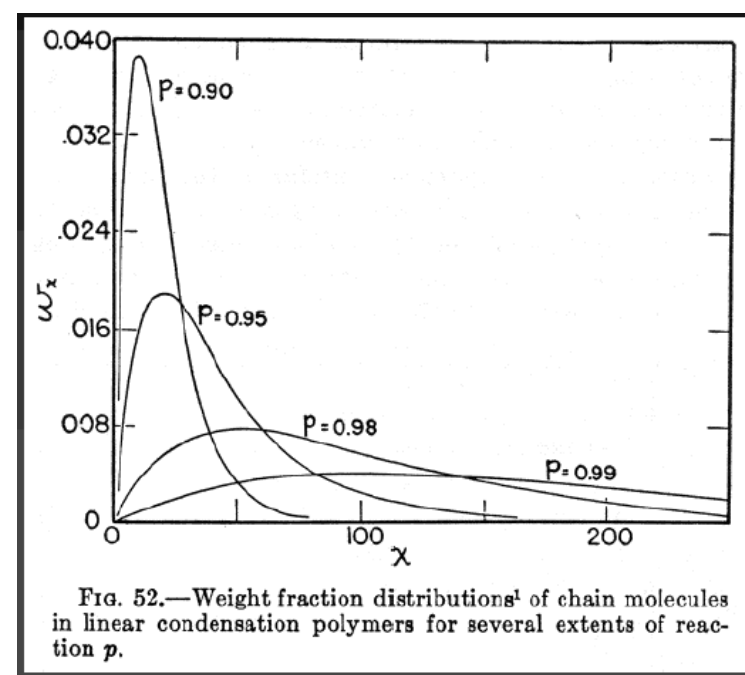
## Most Probable Distribution: Number Fraction

$$N_x = p^{x-1}(1-p)$$



## Most Probable Distribution: Weight Fraction

$$W_x = xp^{x-1}(1-p)^2$$



## 4. Distributions. Statistical growth

### Chain-growth Polymerization

Chain growth polymerization is the formation of polymers from unsaturated monomers.

Consider a chain growing until random something stops it.

Let  $p$  = probability that it keeps on going.

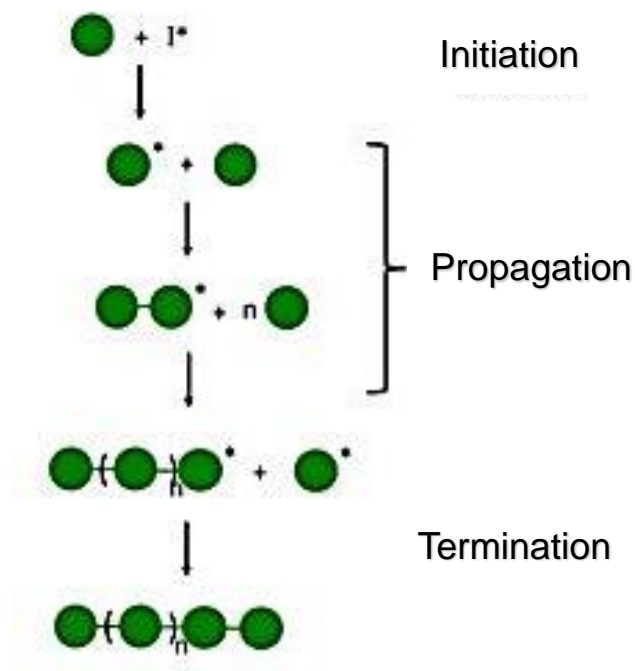
$1 - p$  = probability that it gets stopped

Thus, the probability that an  $x$ -mer forms is given by  $p^{x-1}(1-p)$

According to the initiator used in the chain growth polymerization process, there are three types of chain growth polymerization.

- Radical Polymerization – the initiator is a radical
- Cationic Polymerization – the initiator is an acid/cation
- Anionic Polymerization – the initiator is a nucleophile

### Chain-Growth



## 4. Distributions. Statistical growth

### Chain-growth Polymerization

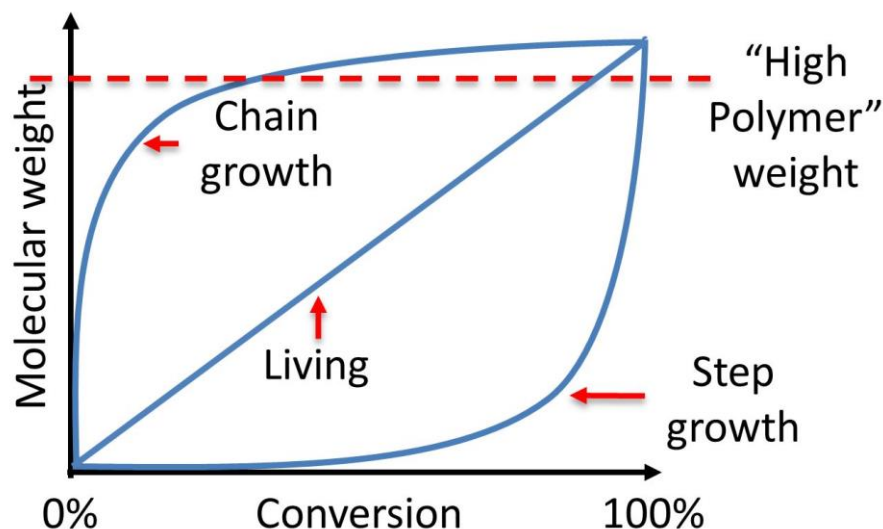
Most Probable Distribution: Number Fraction	$N_x = p^{x-1}(1-p)$
Number Average Molecular Weight, $M_n$	$M_n = \frac{\sum_i N_i M_i}{\sum_i N_i}$
Most Probable Distribution: Weight Fraction	$W_x = xp^{x-1}(1-p)^2$
Weight Average Molecular Weight, $M_w$	$A = \sum_i w_i M_i = M_w$
the Polydispersity	$M_w / M_n$

**It is the same as for step-growth polymerization.** Other things can happen in chain growth polymerization: two growing chains can join their two active ends, etc., **→ it can give rise to different statistics.**

## 4. Distributions. Statistical growth






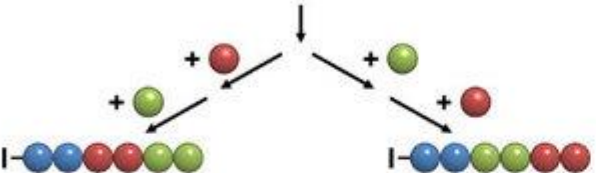
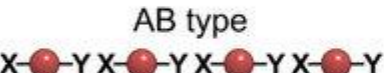
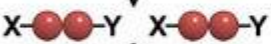

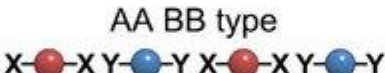
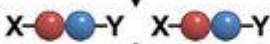


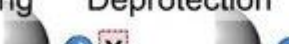

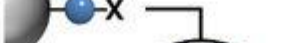

### Differences from chain-growth polymerization

Step-growth polymerization	Chain-growth polymerization
Growth throughout matrix	Growth by addition of monomer only at one end or both ends of chain
Rapid loss of monomer early in the reaction	Some monomer remains even at long reaction times
Similar steps repeated throughout reaction process	Different steps operate at different stages of mechanism (i.e. initiation, propagation, termination, and chain transfer)
Average molecular weight increases slowly at low conversion and high extents of reaction are required to obtain high chain length	Molar mass of backbone chain increases rapidly at early stage and remains approximately the same throughout the polymerization
Ends remain active (no termination)	Chains not active after termination
No initiator necessary	Initiator required

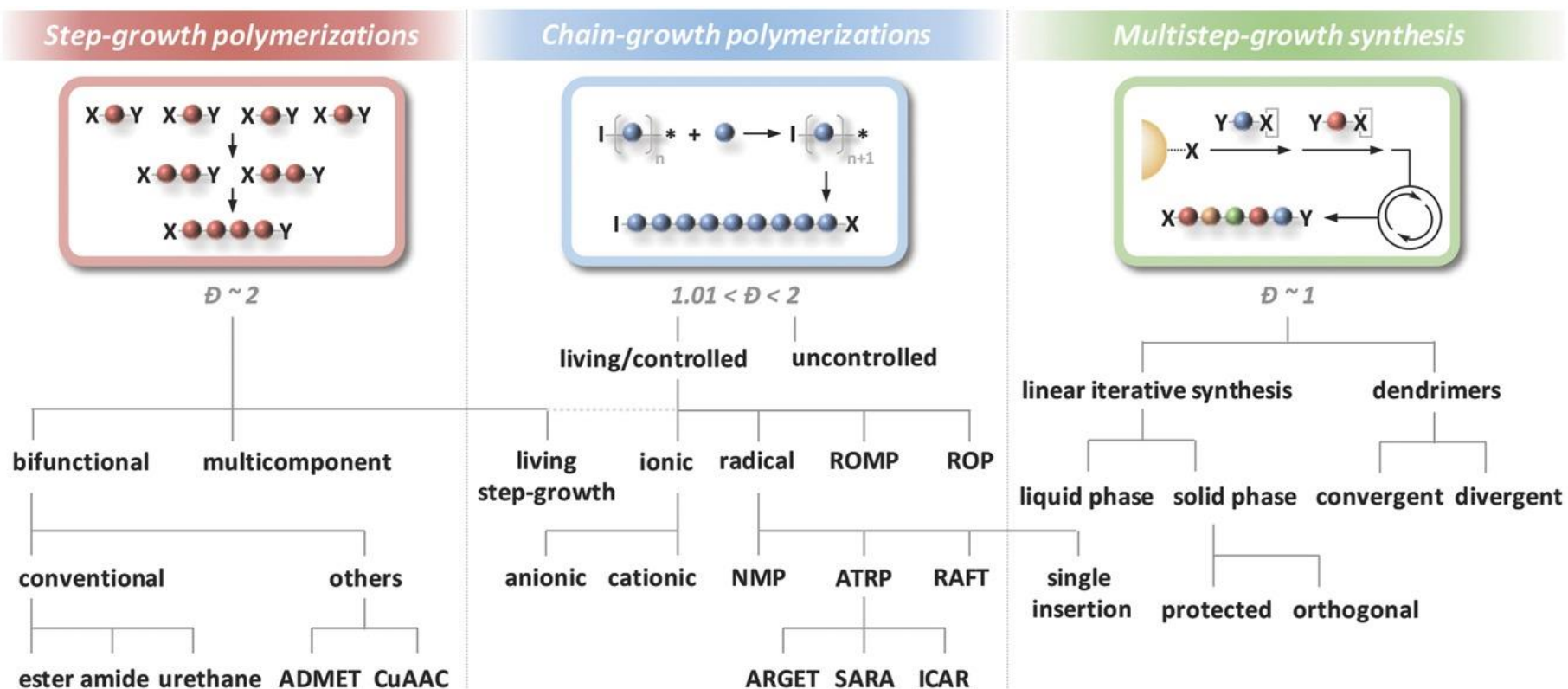




## 4. Distributions. Statistical growth

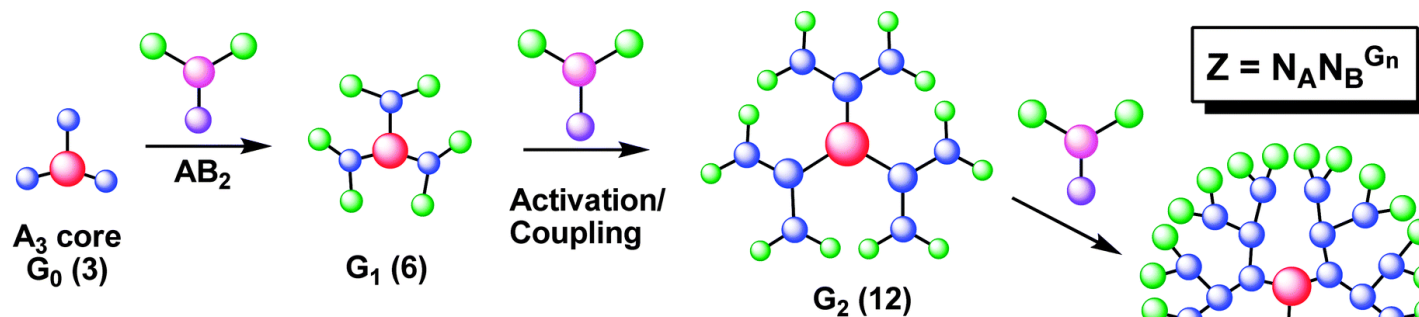
Polymerization method	Accessible molecular weight distribution	Accessible microstructure
<p>Chain-growth polymerization methods</p> <p>Initiator +  → I-[]<sub>n</sub>* +  → I-[]<sub>n+1</sub>* → I-[]</p> 	<p>Broad distribution with uncontrolled methods <math>\bar{M}_w/\bar{M}_n \sim 2</math></p> <p>Narrow distribution with controlled methods <math>\bar{M}_w/\bar{M}_n \sim 1.01</math></p>	<p>Straightforward access of graft and multiblock copolymers</p> <p>Synthesis of gradient, periodic and alternating copolymers is challenging</p>
<p>Step-growth polymerization methods</p> <p>AB type:  →  → </p> <p>AA BB type:  →  → </p>	<p>Generally broad distribution <math>\bar{M}_w/\bar{M}_n \sim 2</math></p>	<p>Straightforward access of alternating, periodic and multiblock copolymers</p>
<p>Multistep-growth coupling</p> <p>Coupling:  → </p> <p>Deprotection:  → </p> 	<p>Monodisperse structures <math>\bar{M}_w/\bar{M}_n \sim 1</math></p>	<p>Complete control over the arrangement of monomers</p>

## 4. Distributions. Statistical growth

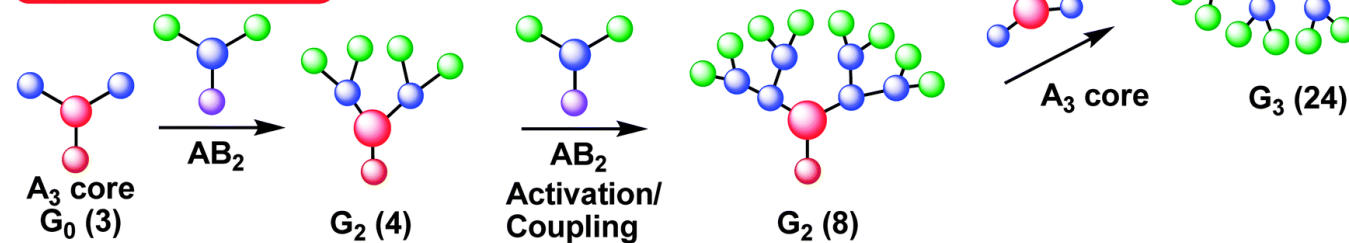


## 4. Distributions. Statistical growth

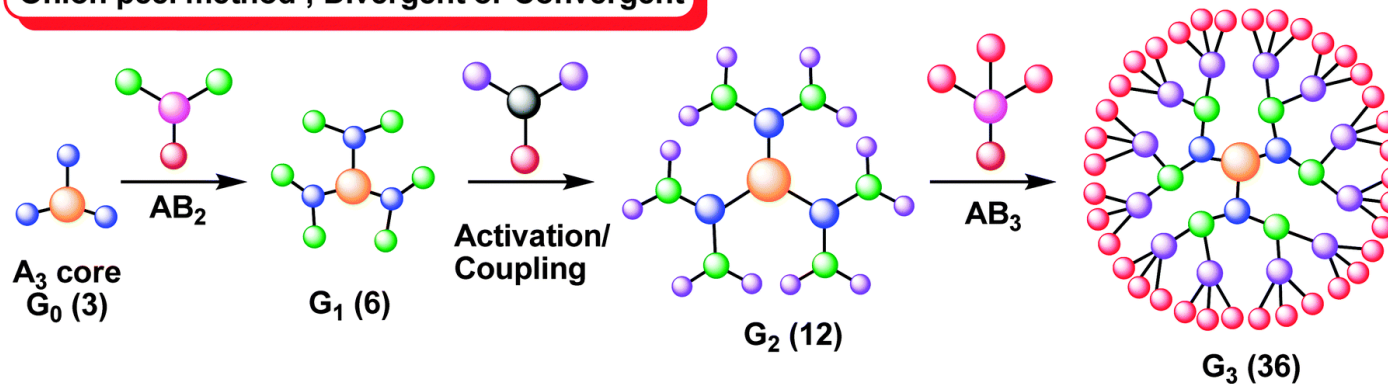
## Divergent method



## Convergent method



## Onion peel method ; Divergent or Convergent



## 5. Molar Distributions

### Molar Mass in Polymers

#### Number average molar weight

$$M_n = \frac{\sum_i c_i}{\sum_i (c_i/M_i)} = \frac{\sum_i N_i M_i}{\sum_i N_i}$$

#### Weight average molar weight

$$M_w = \frac{\sum_i (c_i M_i)}{\sum_i c_i}$$

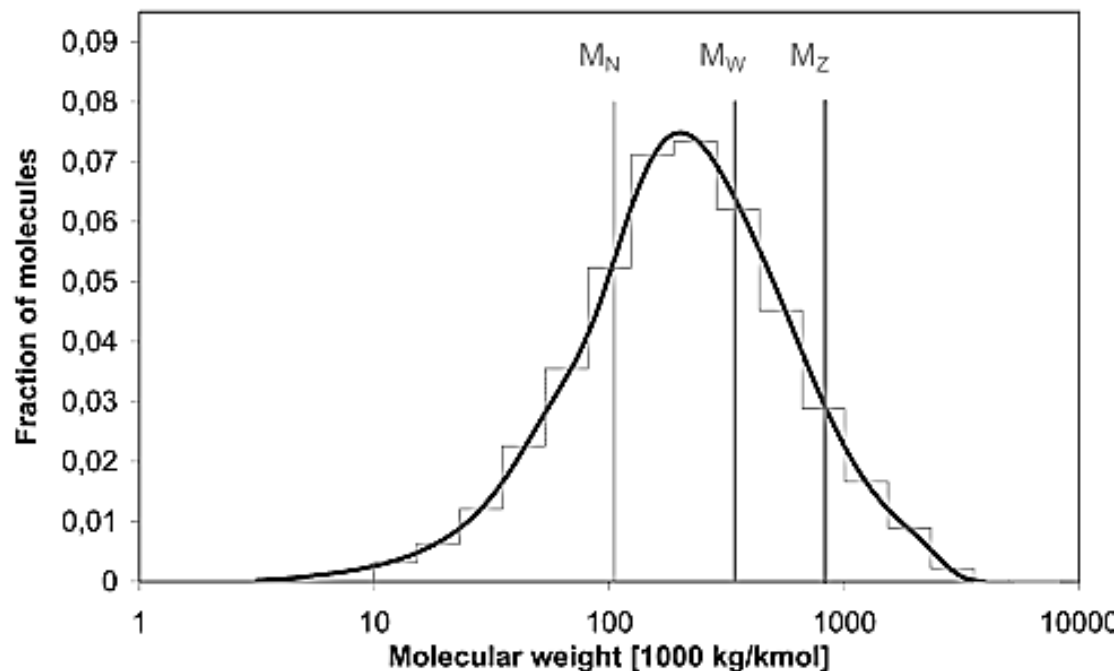
#### z-average molecular weight

$$M_z = \frac{\sum_i (c_i M_i^2)}{\sum_i (c_i M_i)} = \frac{\sum_i z_i M_i}{\sum z_i}$$

**MMD** (polydispersity) is with  $M_w/M_n$

For **monodisperse** samples  $M_w/M_n = 1$

**Polydisperse** polymers have  $M_w/M_n > 1$

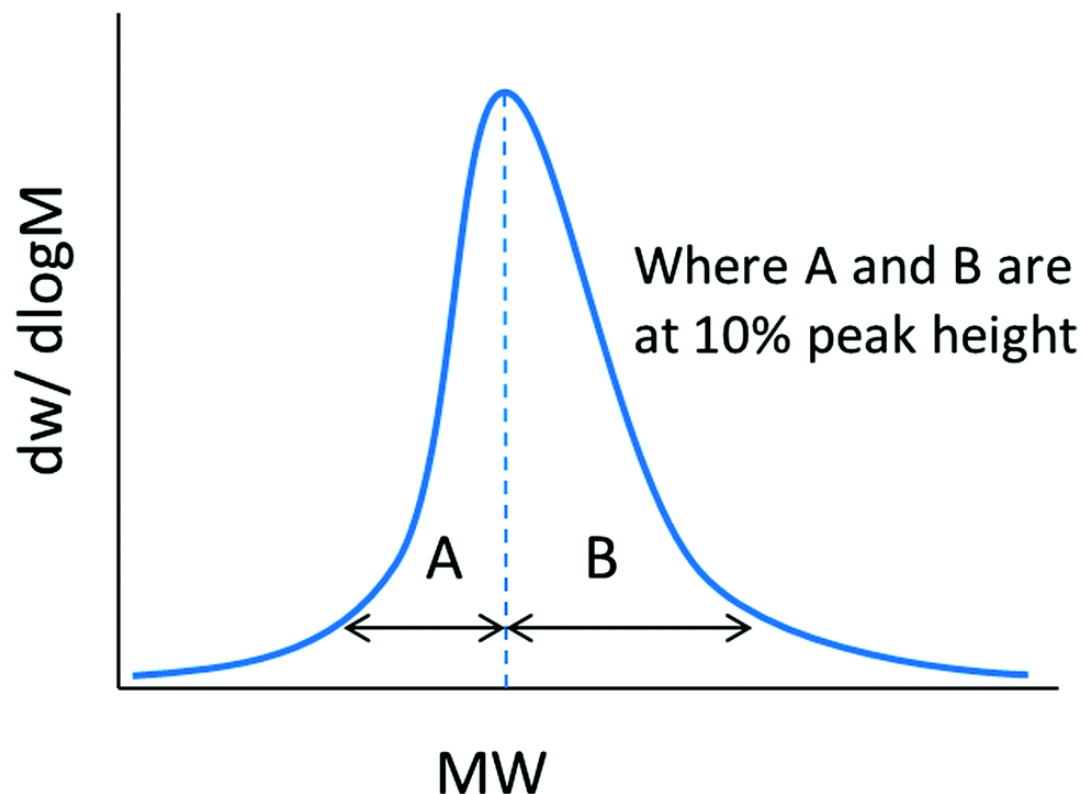


## 5. Molar Distributions

### Symmetry of the shape of the distribution

**Asymmetry factor ( $A_s$ ) to describe the symmetry of the shape of the distribution**, which gives a qualitative measure of the skew of a distribution. This factor is determined as the ratio of the distance from the peak maximum to the front of the peak over the distance from the peak maximum to the back of the peak at 10% of the peak height.

$$A_s = \frac{B}{A} = \left| \frac{MW_{\text{upper 10\%}} - M_p}{MW_{\text{lower 10\%}} - M_p} \right|$$

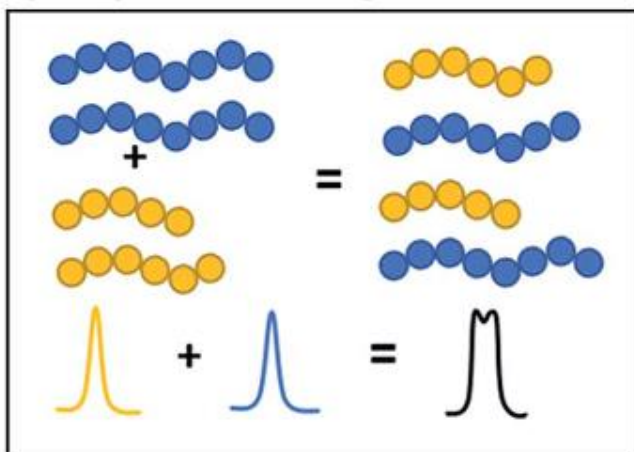




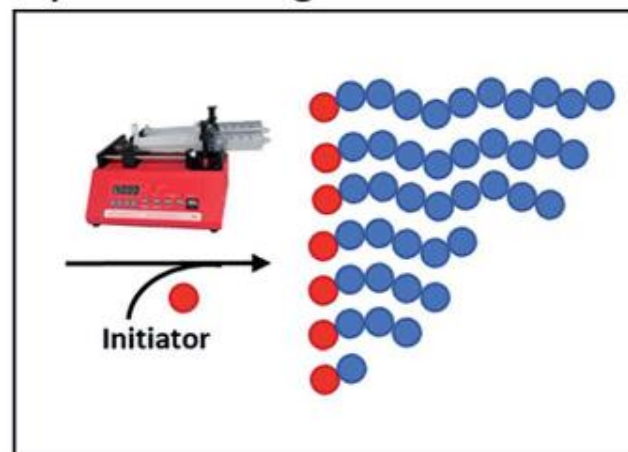
## 6. How can we tune some of this properties?

## Methods to tune dispersity

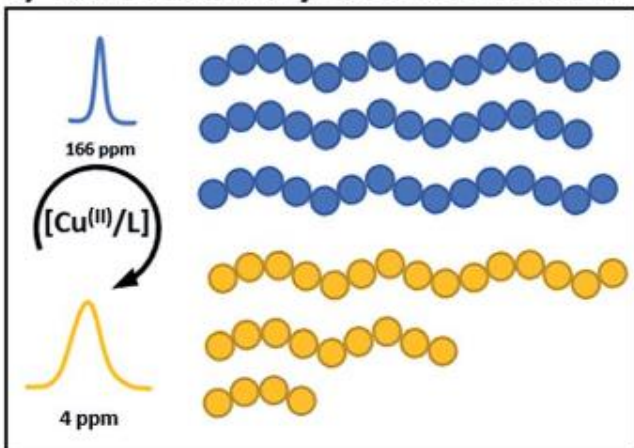
a) Polymer blending



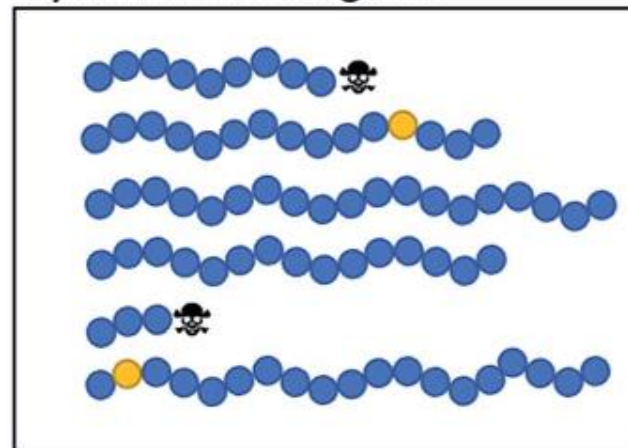
b) Initiation regulation



c) Tailored catalyst concentration



d) Additional reagents



## 6. How can we tune some of this properties?

## Methods to tune dispersity

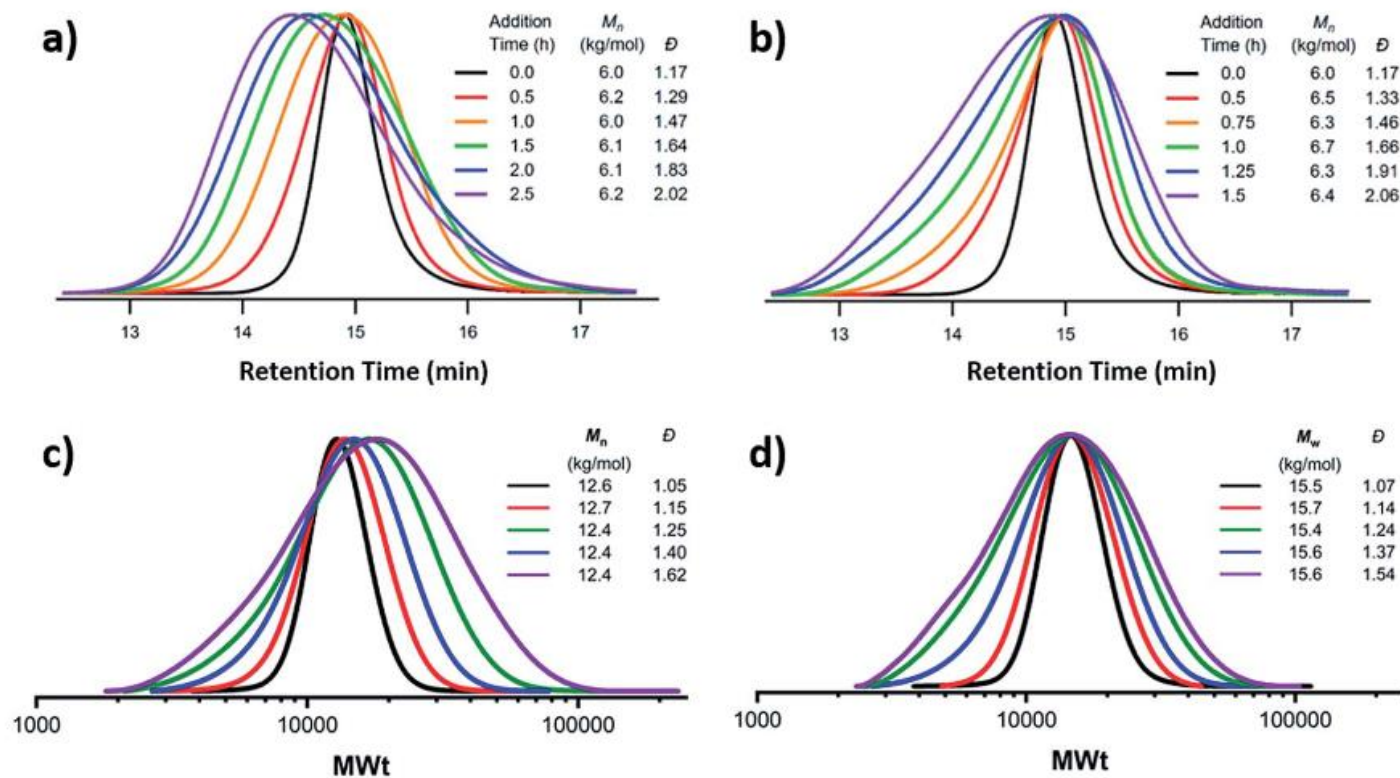
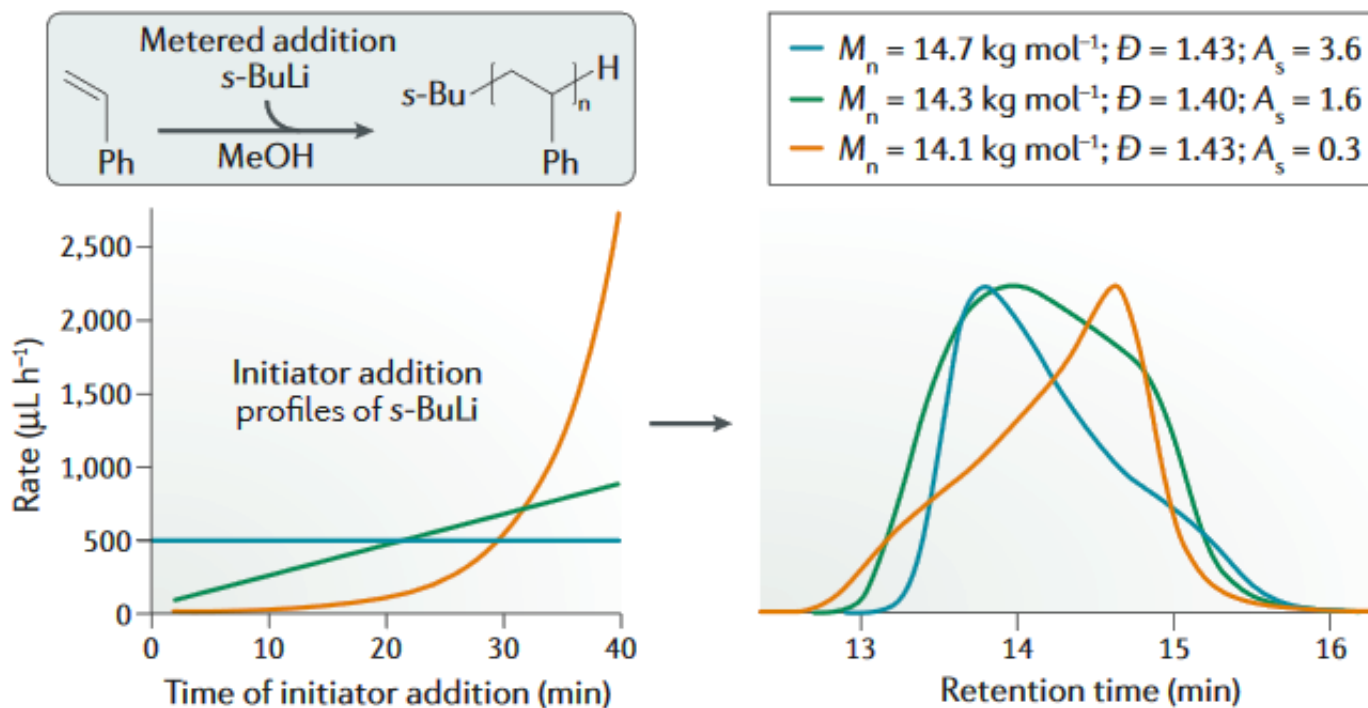


Fig. 4 Comparison of two methods by which the molecular weight distribution is tuned. Firstly, the metered addition of initiator to the NMP of styrene with (a) at a constant rate and (b) with varying rate and secondly via the reduction of catalyst concentration in photo-induced ATRP to maintain (c)  $M_n$  or (d)  $M_w$  constant. This figure is adapted from ref. 53 and 73, permission from ACS publications and Wiley respectively.

## 6. How can we tune some of this properties?

## Methods to tune the shape

## c Metered addition of initiator

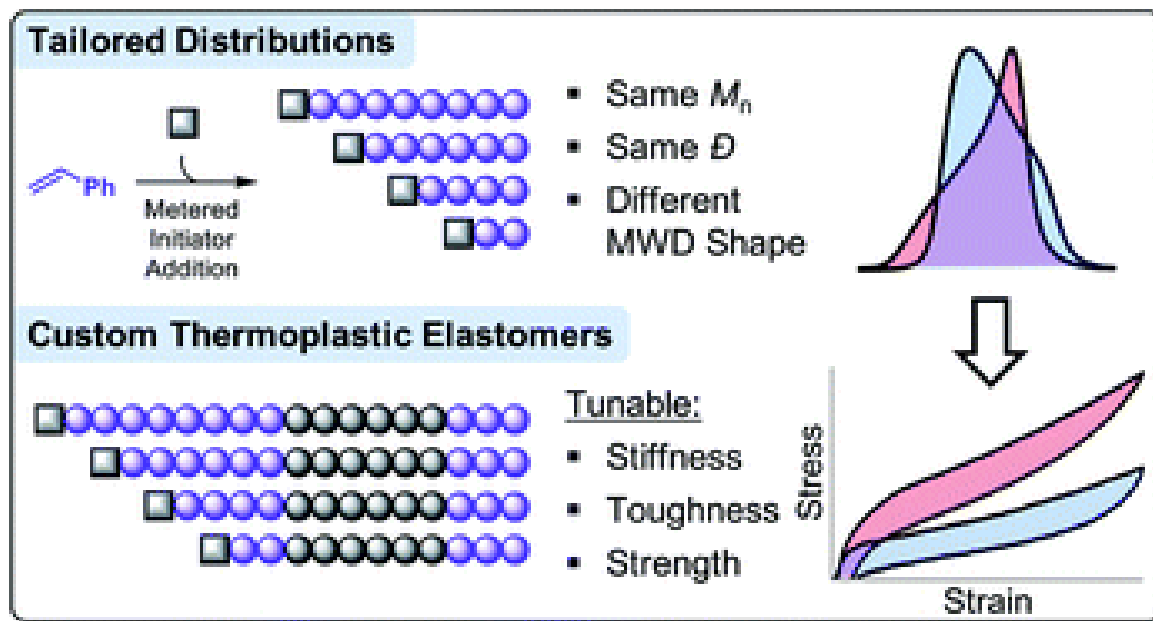




## 6. How can we tune some of this properties?

## Methods to tune the shape

## Tailor-made thermoplastic elastomers: customisable materials via modulation of molecular weight distributions



## 7. Example Questions

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- What is the difference between hyperbranched polymers and dendrimers? How can we determine the degree of branching?
- Define Asymmetry factor ( $A_s$ ). What kind of properties can be influenced?
- Differences from chain-growth polymerization
- How can we tune the polydispersity?